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七 INVESTIGATION OF SPUTTERING EFFECTS ON THE MOON'S SURFACE

Third Quarterly Status Report ? (8 93 08)

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INVESTIGATION OF SPUTTERING EFFECTS ON THE MOON'S SURFACE

Third Quarterly Status Report Contract NASw-751

ABSTRACT

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Experiments to investigate the influence of bombardment of the lunar surface by the solar wind were continued. Darkening of basalt powder surfaces as a function of energy and duration of hydrogen ion bombardment has been determined. Reflection curves for complex solid mineral surfaces before and after bombardment have been measured. The hole-drilling method of determining sputtering yields has been extended to the oxides of Al, Ti, and Fe.

I. INTRODUCTION

The paper, "Modification of the Lunar Surface by the Solar-Wind Bombardment", submitted as the First Quarterly Status Report of this contract has been published in the November issue of Planetary and Space Science. A letter describing our latest research results is being prepared for submission to the Journal of Geophysical Research.

We have continued to survey the literature for solar-wind and lunar surface data. Fielder² has published a paper supporting the importance of intrinsic erosion phenomena on the moon such as seismic and volcanic action. He concludes that intrinsic erosion is more important than extrinsic erosion such as cosmic-ray bombardment and micrometeorite impact. Solar-wind bombardment is not considered a major cause of extrinsic erosion.

Snyder, Neugebauer, and Rao have published a paper on the correlation of solar-wind velocity with cosmic-ray variations and solar and geomagnetic activity. A strong correlation between increased plasma velocity and geomagnetic disturbances suggests the possibility that the abundant geomagnetic data will help settle the question of mean plasma velocity over large numbers of solar activity cycles.

In the third quarter the study of photometric properties of hydrogen-sputtered surfaces was continued. The degree of darkening of basalt powder has been determined as a function of energy and duration of ion bombardment. Normal albedos of various powdered materials before and after bombardment have been measured. The hole-drilling method of determining sputtering yields for metals has been

extended to metal oxides using a technique in which thin oxide foil targets are prepared by pressing and compacting oxide powders.

II. SOLAR-WIND BOMBARDMENT OF BASALT POWDER

As reported in our Second Quarterly Status Report, the reflection curve of a hydrogen bombarded powdered basalt surface closely resembles that of the moon after bombardment times equivalent to \sim 2 x 10^5 years of solar-wind bombardment. We have now studied the situation for shorter bombardment times and measured the change of the reflection curve and the normal albedo as a function of bombardment duration. The samples (insulating) are bombarded with hydrogen ions of average energy 500 eV with a rf method described previously. Figure 1 is a photograph of sputtered basalt samples illustrating the darkening. The approximate equivalent duration of solar-wind action is also shown. The albedo is measured normal to the surface with a phase angle between illuminator and detector of $2\frac{1}{2}$. As photographed, the samples in diffuse light appear as they do to the eye, each one considerably darkened as the duration is increased one order of magnitude, whereas the normal albedo indicates the sharpness of the backscatter peak characteristic of sputtered powder surfaces. Figure 2 is a similar photograph except the samples have just been exposed to the plasma with the target electrically floating. In this case, the target surface charges up negatively and is bombarded by ions of much lower ion energy (probably not exceeding 100 eV). The ion current density however remains the same. We have no data on hydrogen sputtering yields at such low bombarding

DARKENING OF BASALT POWDER BY SIMULATED SOLAR-WIND BOMBARDMENT

| Sputtering Duration | | Normal Albedo |
|--|--|---------------|
| original | | ~ 0.27 |
| 3 1/3 minutes (100 YR. EQUIV.) | | 0.16 |
| 33 minutes (103 YR. EQUIV.) | | 0.07 |
| 5½ hours (10⁴ YR. EQUIV.) | | 0.05 |
| 55 hours (10 ⁵ YR. EQUIV.) | | 0.06 |

FIGURE 1 - Darkening of Basalt Powder as a Function of Hydrogen Ion Bombardment

DARKENING OF BASALT POWDER BY HYDROGEN PLASMA EXPOSURE

| Exposure Duration | | Normal Albedo | |
|--------------------------|--|---------------|--|
| original | | ~ 0.27 | |
| 3 ^{1/3} minutes | | 0.25 | |
| 33 minutes | | 0.18 | |
| 5½ hours | | 0.12 | |
| 55 hours | | 0.09 | |

FIGURE 2 - Darkening of Basalt Powder as a Function of Hydrogen Plasma Exposure

energies and therefore cannot correlate these results with actual solarwind bombardment times. Physical sputtering at such low energies is probably negligible in the case of hydrogen and we believe that the effects
here are more of a chemical or reactive nature. Experiments with powdered and solid surfaces, where a part of the surface is covered with
a thin quartz foil, indicate that the darkening is certainly not due
to ultraviolet radiation of wavelengths transmitted by quartz. Darkening by exposure to a plasma in noble gases is indeed very slight,
or negligible, indicating the importance of chemical effects in the
case of hydrogen.

Figures 3-6 are reflection curves of visible light for the surfaces illustrated in Figures 1 and 2. The angles of incidence i and reflection ϵ are measured from a normal to the surface. The backscatter is more pronounced on the sputtered surfaces than on the plasma-exposed surfaces. The curve for the equivalent of $\sim 2 \times 10^5$ years solarwind action does not differ significantly from that for 10^5 years equivalent; therefore, we feel the surface has reached an equilibrium state.

III. SOLAR-WIND BOMBARDMENT OF SOLID SCORIA AND PUMICE

According to Hapke, 4 a complex, underdense surface is required of a material for it to have reflection properties similar to those of the moon. In an attempt to determine the effects of sputtering on solid but complex surfaces, flat scoria and pumice surfaces were bombarded by hydrogen ions for a period equivalent to 10⁴ years solar-wind action. The reflection curves of these surfaces before and after bombardment are shown in Figures 7-10. Both surfaces are darkened by sputtering,

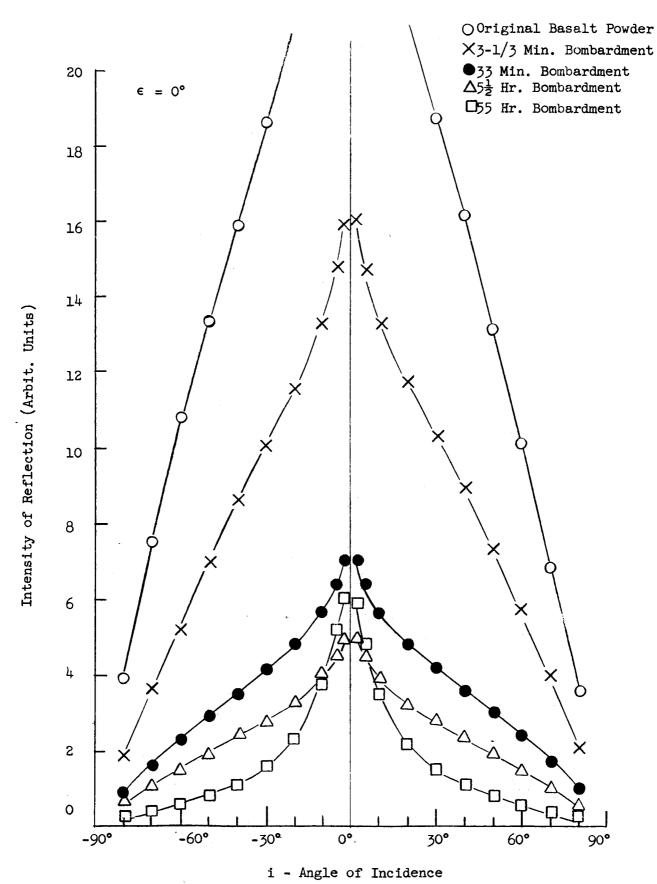


FIGURE 3 - Reflection Curves of Basalt Powder as a Function of Hydrogen Ion Bombardment. $\epsilon = 0^{\circ}$.

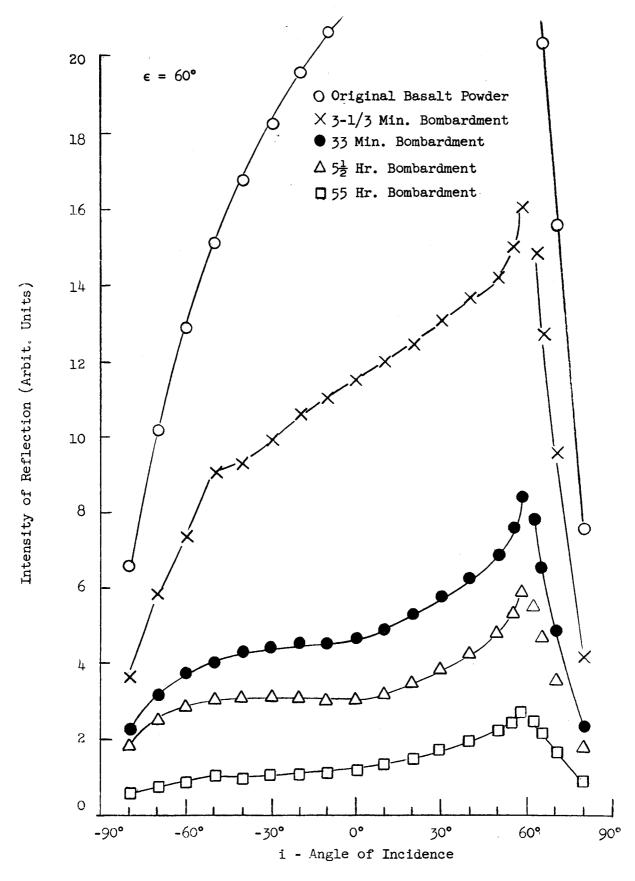


FIGURE 4 - Reflection Curves of Basalt Powder as a Function of Hydrogen Ion Bombardment. $\epsilon = 60^{\circ}$.

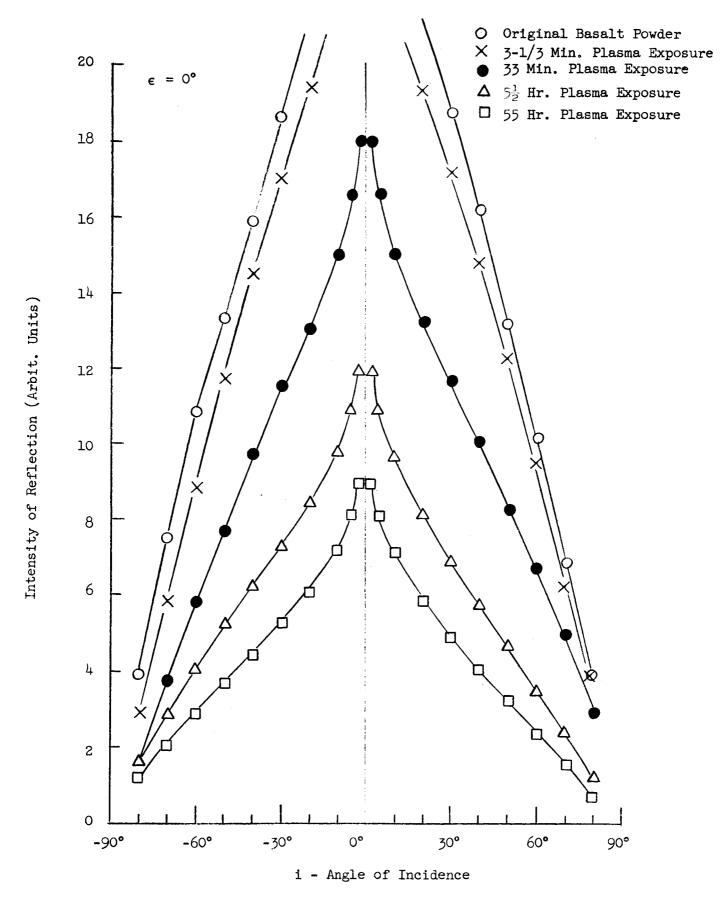


FIGURE 5 - Reflection Curves of Basalt Powder as a Function of Hydrogen Plasma Exposure. $\epsilon = 0^{\circ}$.

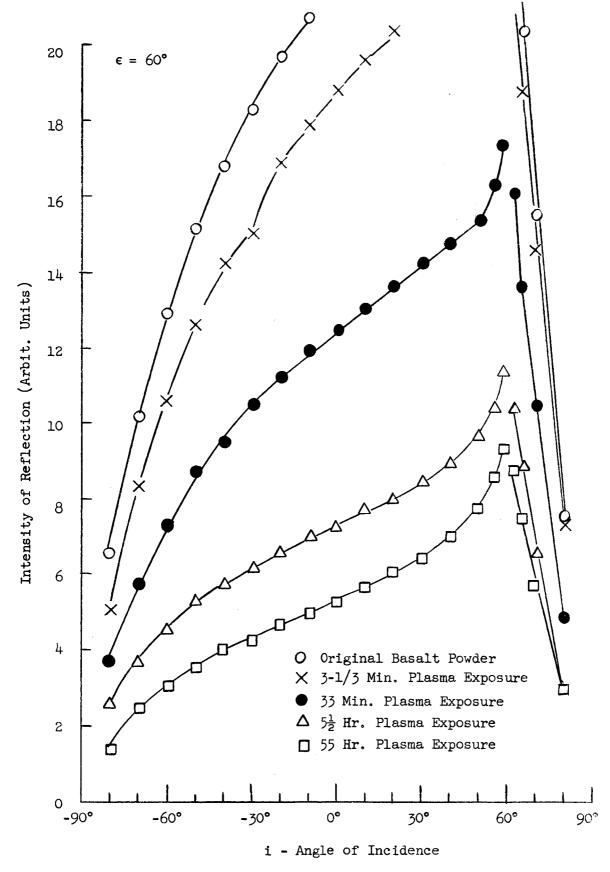
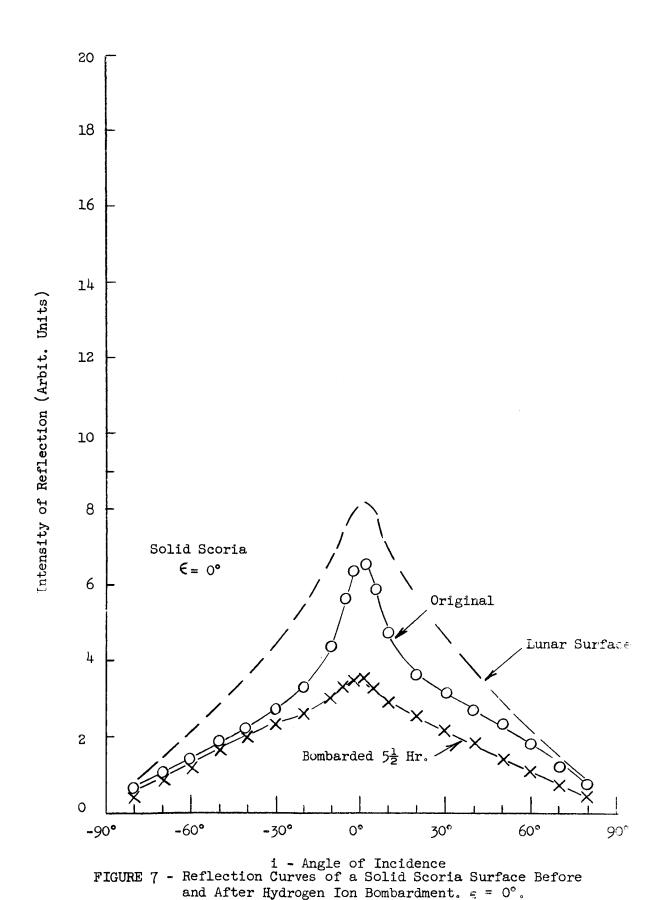


FIGURE 6 - Reflection Curves of Basalt Powder as a Function of Hydrogen Plasma Exposure. $\epsilon = 60^{\circ}$.



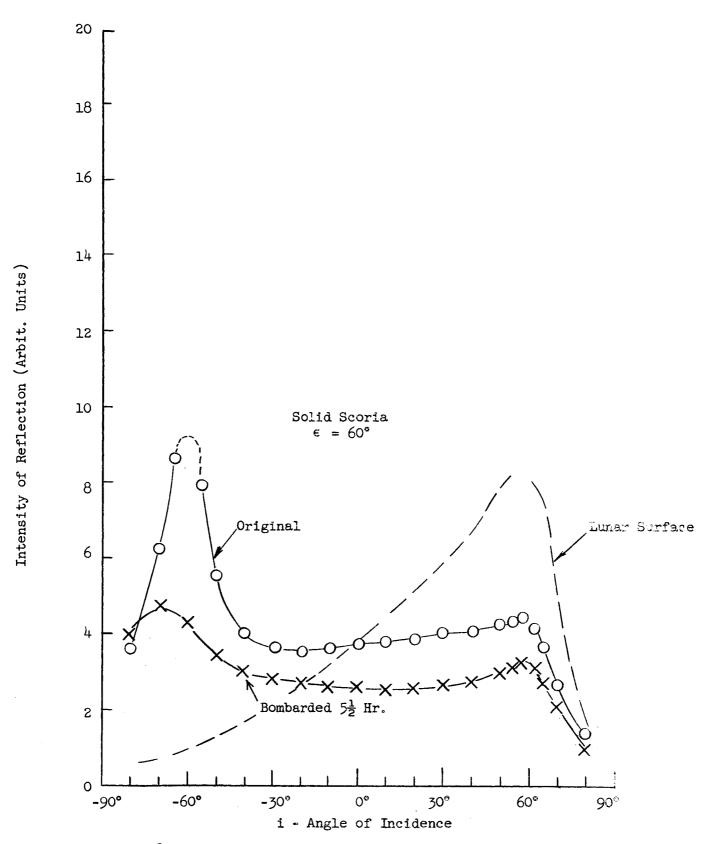


FIGURE 8 - Reflection Curves of a Solid Scoria Surface Before and After Hydrogen Ion Bombardment. $\epsilon=60^{\circ}$.

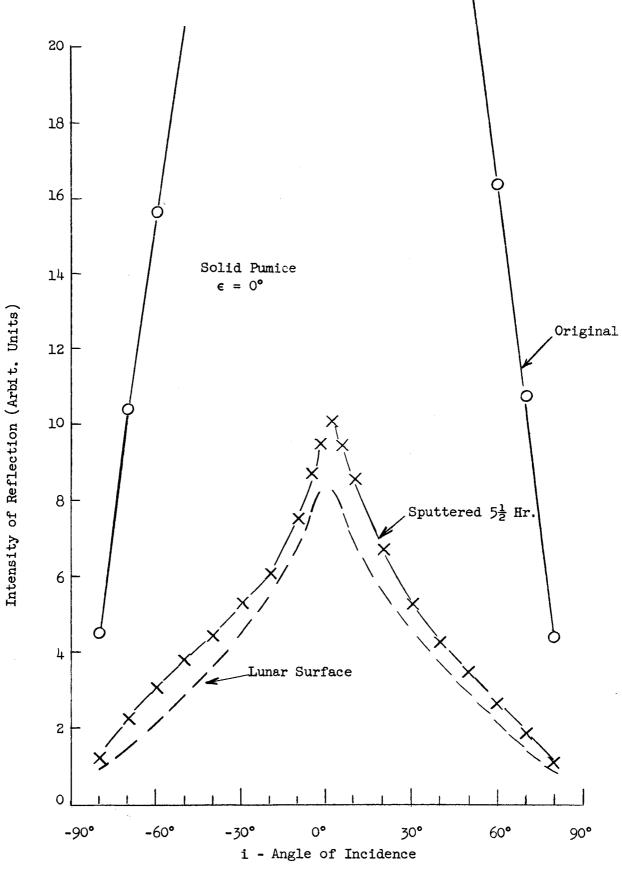


FIGURE 9 - Reflection Curves of a Solid Pumice Surface Before and After Hydrogen Ion Bombardment. ϵ = 0°.

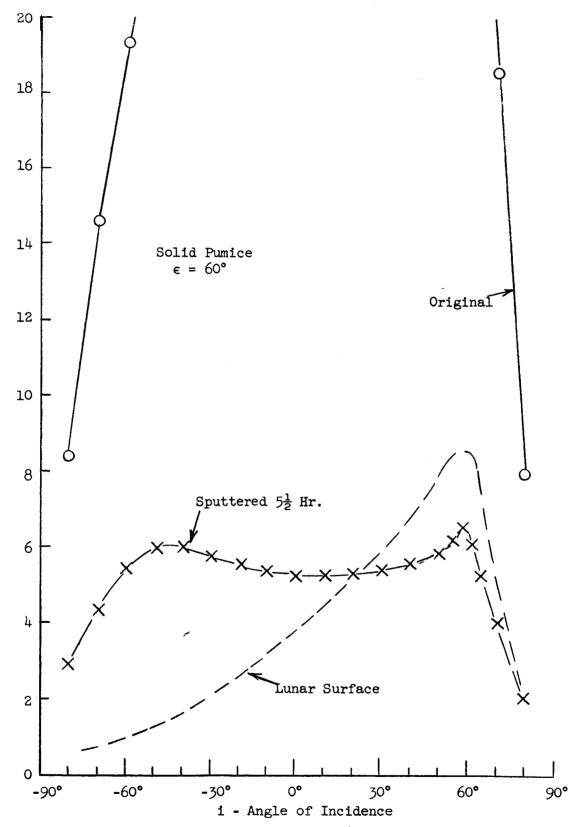


FIGURE 10 - Reflection Curves of a Solid Pumice Surface Before and After Hydrogen Ion Bombardment. $\epsilon = 60^{\circ}$.

the normal albedo being reduced from 0.065 to 0.035 for scoria and 0.37 to 0.10 for pumice. With scoria, the most pronounced effect of hydrogen bombardment is the reduction of the specular reflection from the surface as indicated in the reduction of the backscatter peak at $i = 0^{\circ}$ in Figure 7 and in the $i = -60^{\circ}$ peak in Figure 8. The specular reflection peaks are still pronounced with both surfaces, but the changes in reflection characteristics due to sputtering tend to approach more the reflection curves of the moon.

As has been stated before, we believe the darkening is a combination of physical and chemical effects. We have shown that simulated solar-wind action on many terrestrial materials results in a darkened surface of neutral color with pronounced backscatter.

IV. SPUTTERING YIELDS OF METAL OXIDES FOR HYDROGEN IONS

The hole-drilling method previously used to obtain sputtering yields of metals with mass-analyzed hydrogen beams has been extended to the oxides of metals by the development of techniques for preparing oxide foils.

Since the composition and area density of the foil must be known, the technique must permit determination of each quantity. The method of baking at high temperatures in the presence of oxygen is adequate for metals of high melting point and relatively high chemical activity. Thus Ti and Fe foils were cut to size, placed on quartz, and baked in air for several hours at 1000°C. Some difficulty was experienced with curling of the foils and sticking to the quartz, but those foils that curled such that only the corners stuck to the quartz could be broken

loose and used. The oxygen content was obtained from the increase in area density. The foils became essentially ${\rm TiO}_2$ and ${\rm Fe}_2{\rm O}_3$, the highest oxidation states. The latter substance is non-magnetic as was the foil.

The second method is the compacting of fine powders of a known substance, in this case $\mathrm{Al}_2\mathrm{O}_3$ and $\mathrm{Fe}_2\mathrm{O}_3$, into a self-supporting foil. The powders were spread thinly, and as uniformly as possible on a Ag foil, a 4 x 10 mm rectangle. The foil and powder were compressed between accurately parallel, polished steel surfaces to an estimated 16,000 atmospheres. The resultant powder foil was self-supporting over the 4 x 10 mm rectangle, while the spilled-over powder could be brushed away from its edge. The strongly-compressed rectangle of powder had undergone a marked color change to darker shades and its surface was shiny. Furthermore, it readily separated from the Ag foil. The use of a Ni foil to define the area of high pressure was not satisfactory because the powder foil tended to show numerous cracks leading in from the edge toward the central region. The softer Ag foil produced few such cracks in the powder foil. Both the Ni and Ag foils were about 25 μ thick.

A calculable thickness of Ni (about $9\,\mu$) was sputtered on one side of the oxide foils to provide additional strength and a path for heat and electrical current from the ion-bombarded area of the foil to its edge. The Ni film faces away from the bombarding mass-analyzed hydrogen ion beams so that it will not be sputtered away from the edge of the hole being drilled. The Ni film is

thick enough to keep the foil temperature at the hole below 500°C; thermal fracture has not been observed.

Foils of quartz have been previously studied by cutting fragments from a large bubble of quartz. But silicon metal could not be obtained as a foil. It is expected that both Si and SiO₂ can be prepared by the powder foil method so that a direct comparison can be made and the role of chemical effects elucidated.

The results are presented in Table I. It should be noted that for each run the ratio $Y_3:Y_2\cong 3:2$. This result is a general feature for this energy range of hydrogen sputtering and can be used as a rough check on the uniformity in thickness of the powder foils. Secondly, the oxide yields are decidedly reduced for Al (x.3), Ti (x.7), and Fe (x.4) when expressed as metal atom/hydrogen ion as they are given in the table. We have observed that metal oxides have yields that are lower than the corresponding metals by an order of magnitude when bombarded by low-energy, noble-gas ions. The fact that the reductions here are rather nearer one half suggests the possibility of chemical removal of oxygen. But oxygen is not removed so very readily that its effect is negligible. The correlation to the heats of formation of the respective oxides, as given in Table II, is not obvious. Results are being repeated for verification.

TABLE I SPUTTERING YIELDS FOR 7 keV $\rm H_2^+$ AND $\rm H_3^+$ IONS

| Preparation | Powder Powder | Baked | Baked Powder Foil |
|----------------------------------|------------------------|-------------------|--|
| $r_3 \times 10^3$ | 15.0 ±1.5 17.1 ±1 | 12.9± 0.6 | 20.4 ±1 12.8 ±1 34 ± 3 |
| $r_2 \times 10^5$ | 10.5 ± 1 11.8 ± 0.5 | 7.5 ± 0.4 | 12.7 ± 0.6 9.4 ± 1 19 ± 2 |
| Oxide | A10 _{3/2} | T_{10_2} | $\frac{\text{FeO}_3/2}{\text{FeO}_3/2}$ S1O_2 |
| Y ₅ × 10 ⁵ | 42 ± 1.5 | 15.3±0.5 | 33 ± 1.5 |
| $\frac{Y_2 \times 10^3}{}$ | 29 ± 1 | 10.9± 0.3 | 24 ± 1 |
| Metal | ΑΊ | Ħ | ਜ e |

TABLE II

HEATS OF FORMATION

| A10 _{3/2} | 199.5 kcal/mole |
|--------------------|-----------------|
| TiO2 | 218.0 |
| FeO _{3/2} | 98.2 |
| SiO2 | 205.4 |

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